

# Fragility and the Rate of Structural Ordering in a Supercooled Metallic Liquids

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National Science Foundation  
WHERE DISCOVERIES BEGIN



# Metallic Glasses

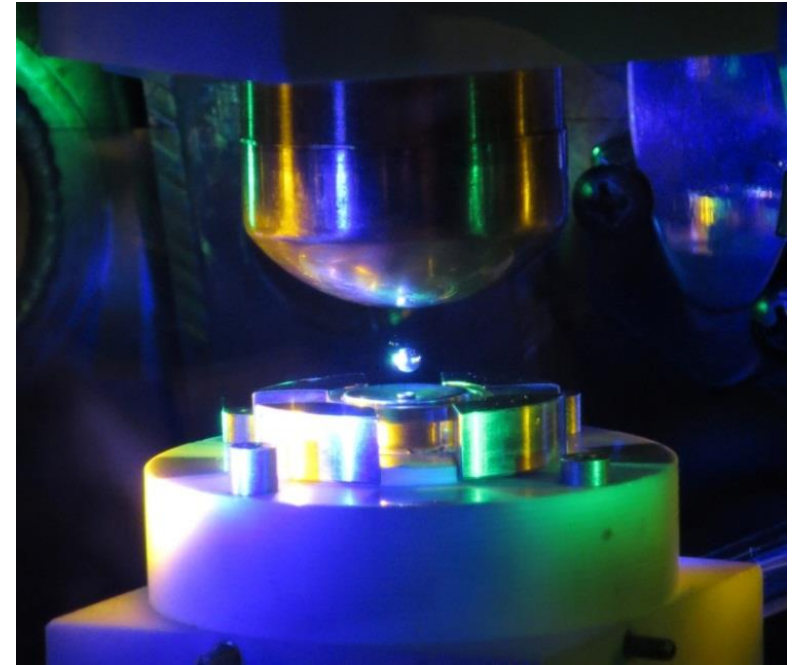
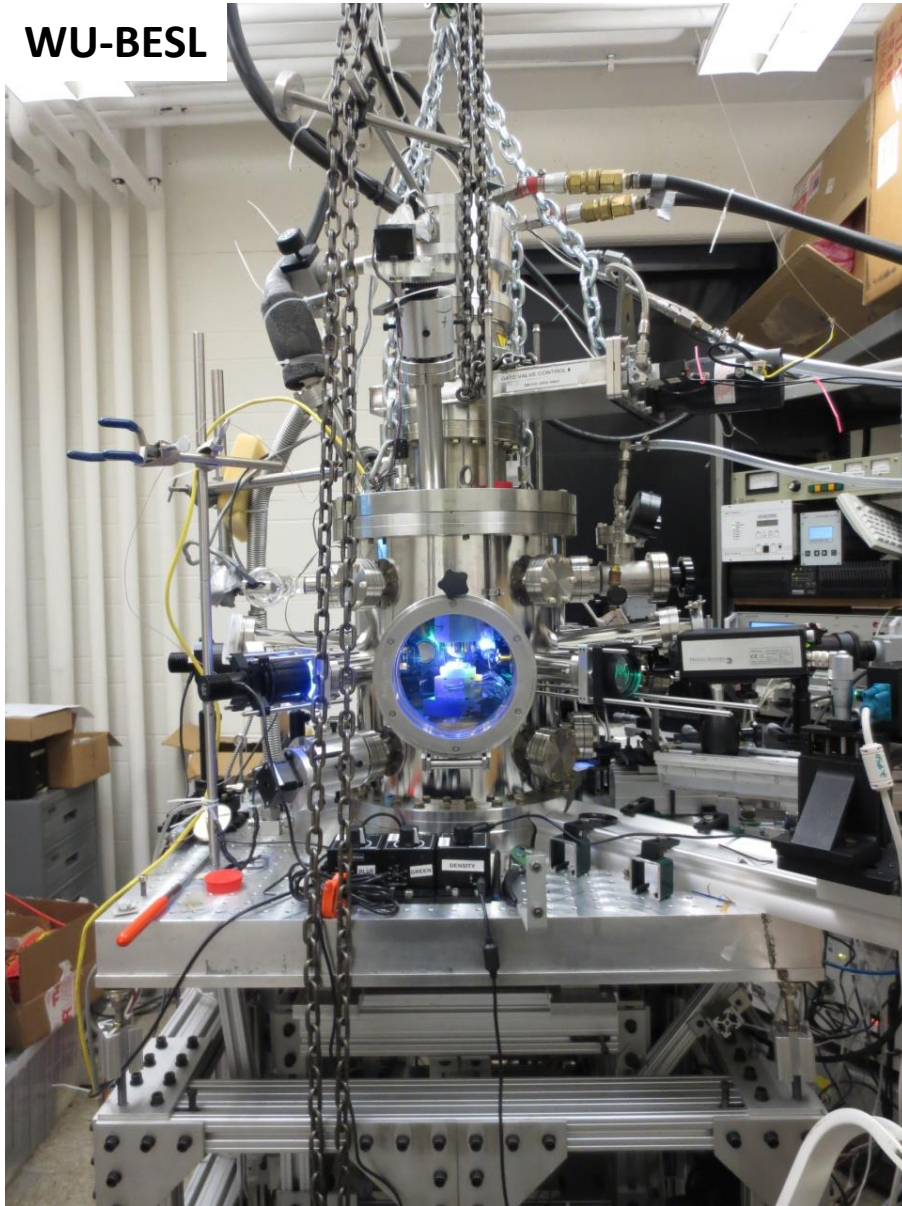


Are There Clues for Good Glass Formation in the High Temperature Liquid?



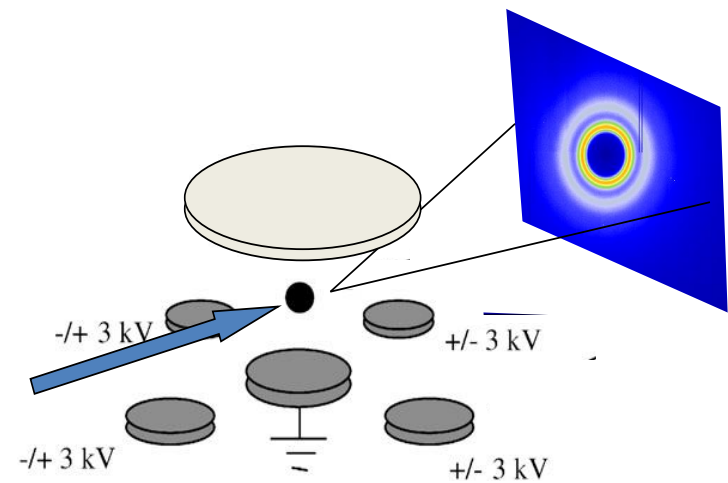
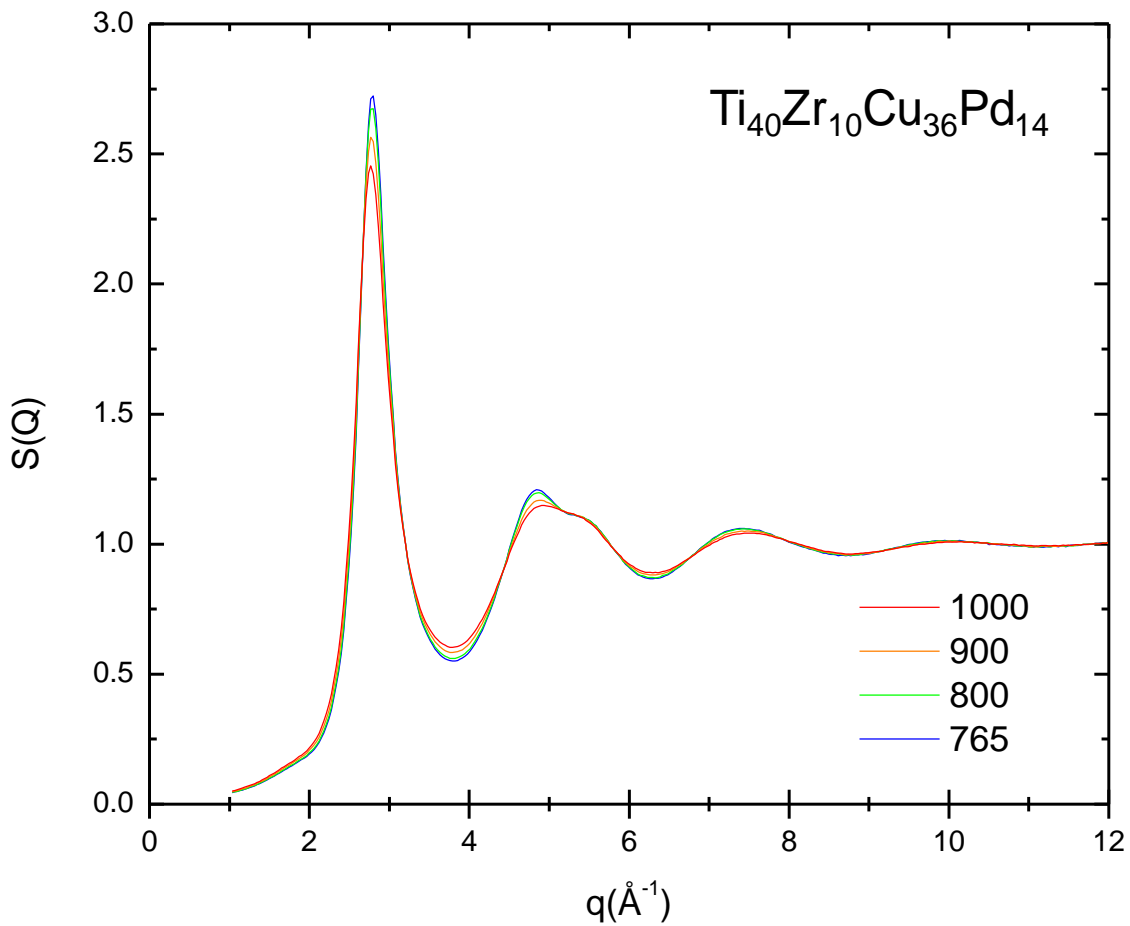
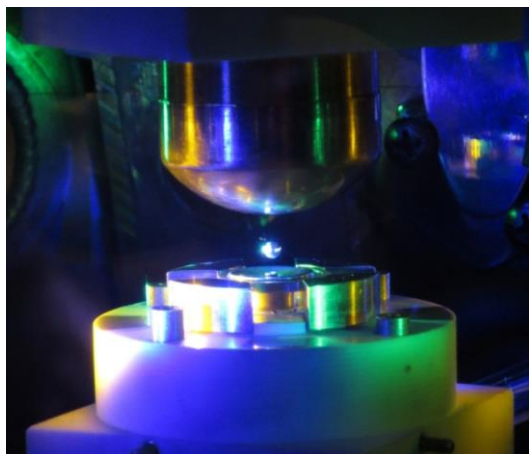
# Containerless Processing – ESL (Electrostatic Levitation)

WU-BESL

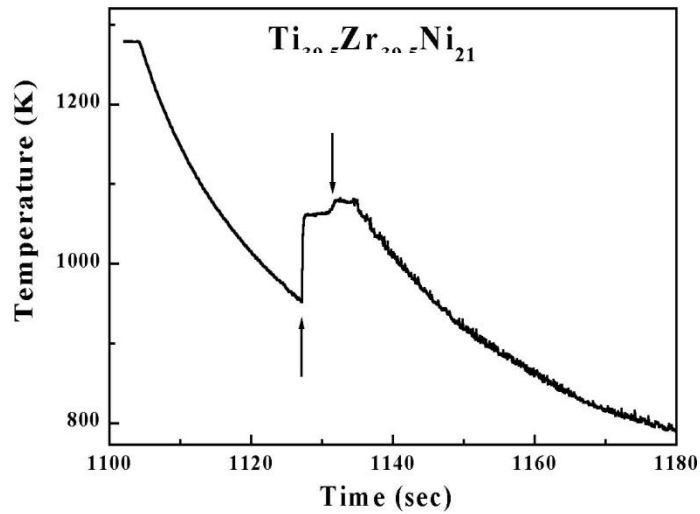


- Designed to be transportable
- Liquid measurements
  - Synchrotron X-ray scattering
  - Maximum undercooling
  - Specific heat
  - Density
  - Viscosity
  - Surface tension

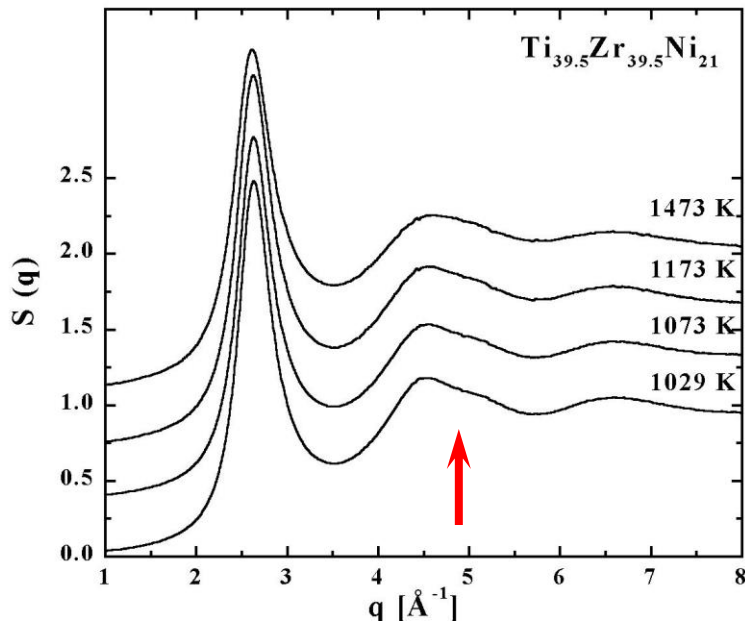
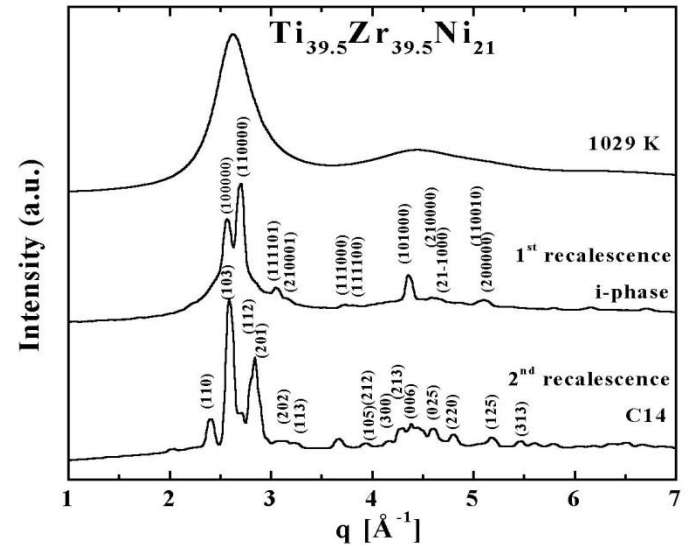
# X-ray Scattering Studies with WU-BESL



# Icosahedral Ordering in a $\text{Ti}_{39.5}\text{Zr}_{39.5}\text{Ni}_{21}$ Liquid and Nucleation



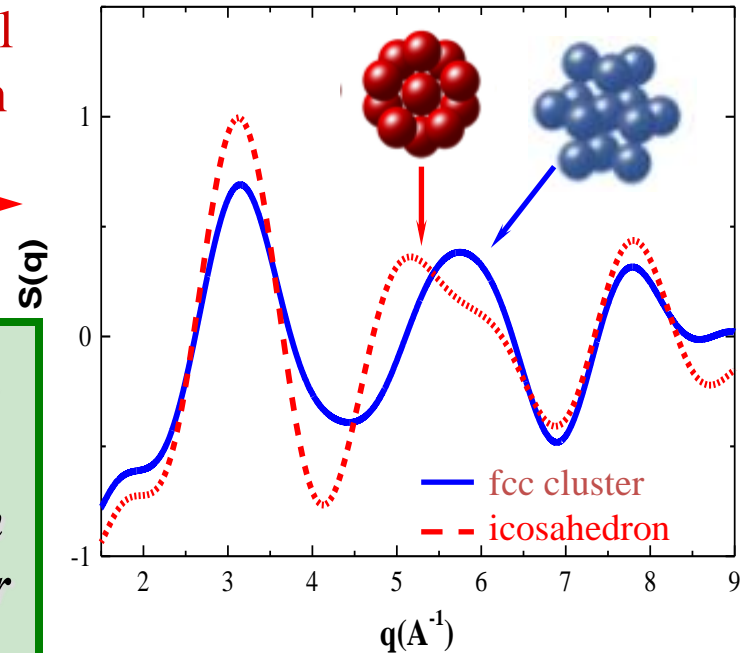
Recalescence  
to metastable  
quasicrystal



Icosahedral  
ordering in  
liquid



Ordering  
lowers  
nucleation  
barrier for  
i-phase



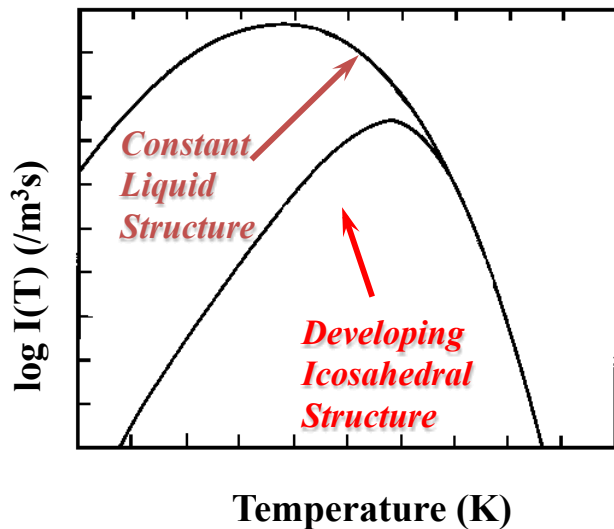
# Consequences of ISRO

## Crystal Nucleation Rate

$$I = \frac{A}{h} \exp\left(-\frac{B}{T} \frac{S^3}{Dg^2}\right)$$

Based on Experiment and MD, ISRO:

- Increases  $\sigma$
- Lowers energy of liquid, decreasing  $\Delta g$
- Increases viscosity

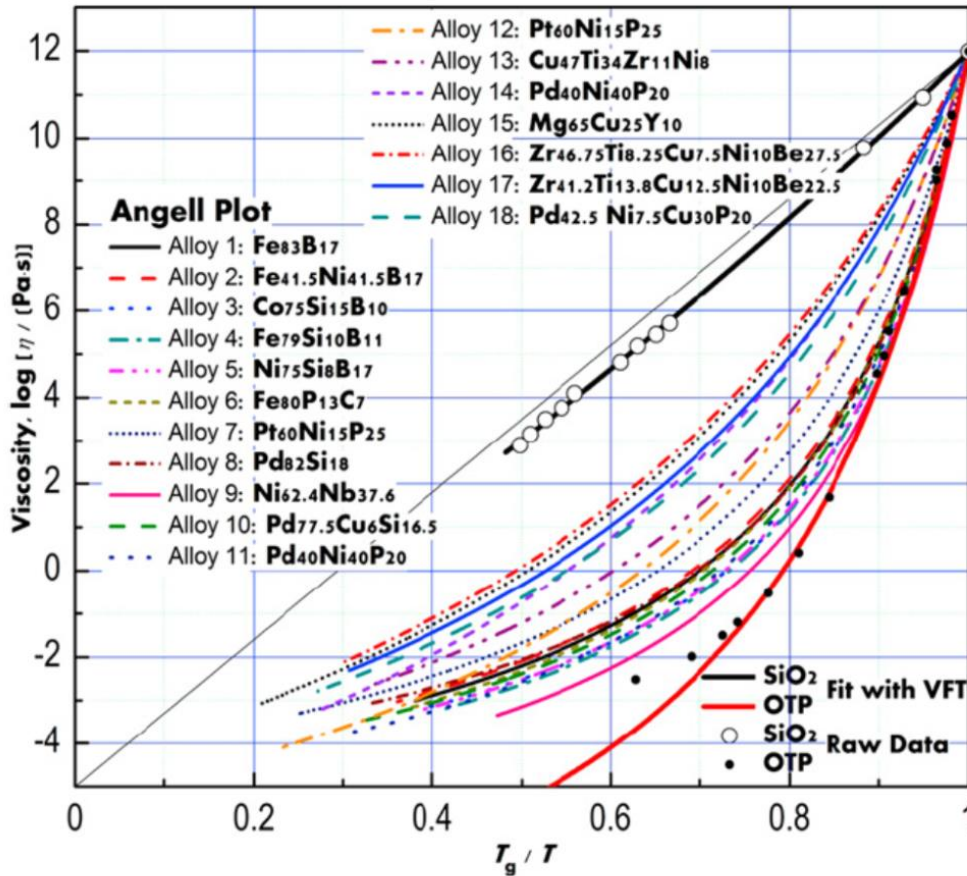


- Decreases nucleation rate for crystal phases

**But:**

- While icosahedral ordering is very common in transition metal liquids, other alloy liquids likely have different local local ordering structures.***
- Other types of local order in liquids, if they lower the energy of the liquid and are incompatible with structures of potential crystal products, will help glass formation in the same way as icosahedral ordering in transition metal liquids.***
- Icosahedral ordering can actually HURT glass formation, increasing the chance for icosahedral phase formation.***

# Liquid Fragility and Glass Formation



T<sub>g</sub>

1446

1112

713

670

550

370.5

yCl 275

332

300

85.2

bonate 152

239.7

- Fragility Index,  $m$

$$m = \left. \frac{d(\log \eta)}{d(T_g / T)} \right|_{T=T_g}$$

- Larger  $m$  – fragile
- Smaller  $m$  – strong

A. Takeuchi, H. Kato, A. Inoue, *Intermetallics*, **18**, 406-411 (2010)

Stronger liquids  
argued to be better  
Metallic glass formers

R. Busch et al., *Acta Mater.* **46**, 4725 (1998).

Sha et al, *JAP*, **105**, 043521 (2009)

Russew et al, *J. Phys.: Conf. Ser.*, **144**, 012094 (2009)

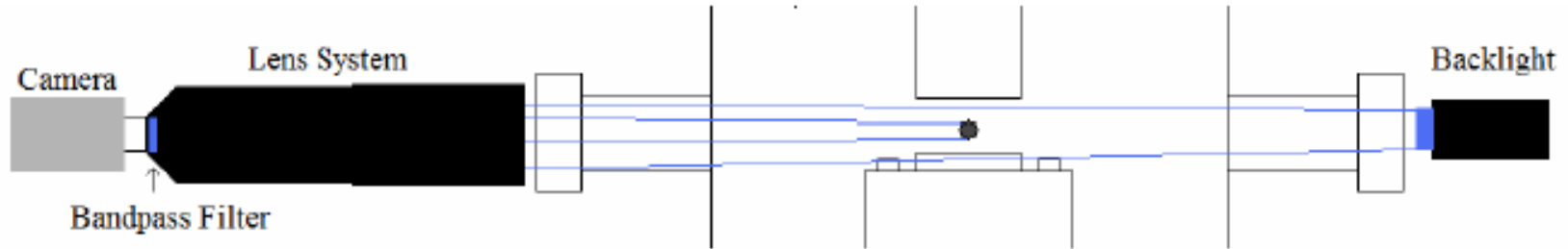
Jakse and Pasturel, *Phys. Rev. B*, **78**, 214204 (2008)

**Is there a Structural  
Signature of  
Fragility?**



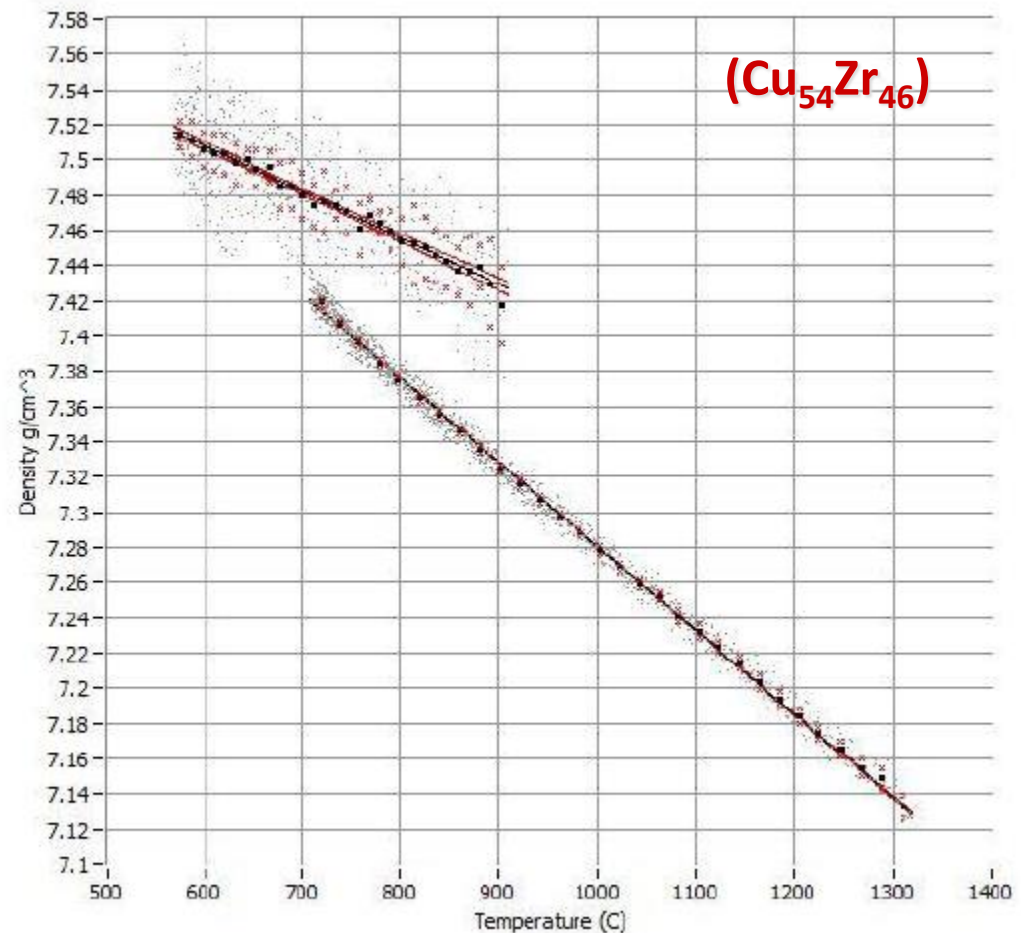
***Temperature  
Dependence  
Of the Volume***

# Density Measurements in BESL

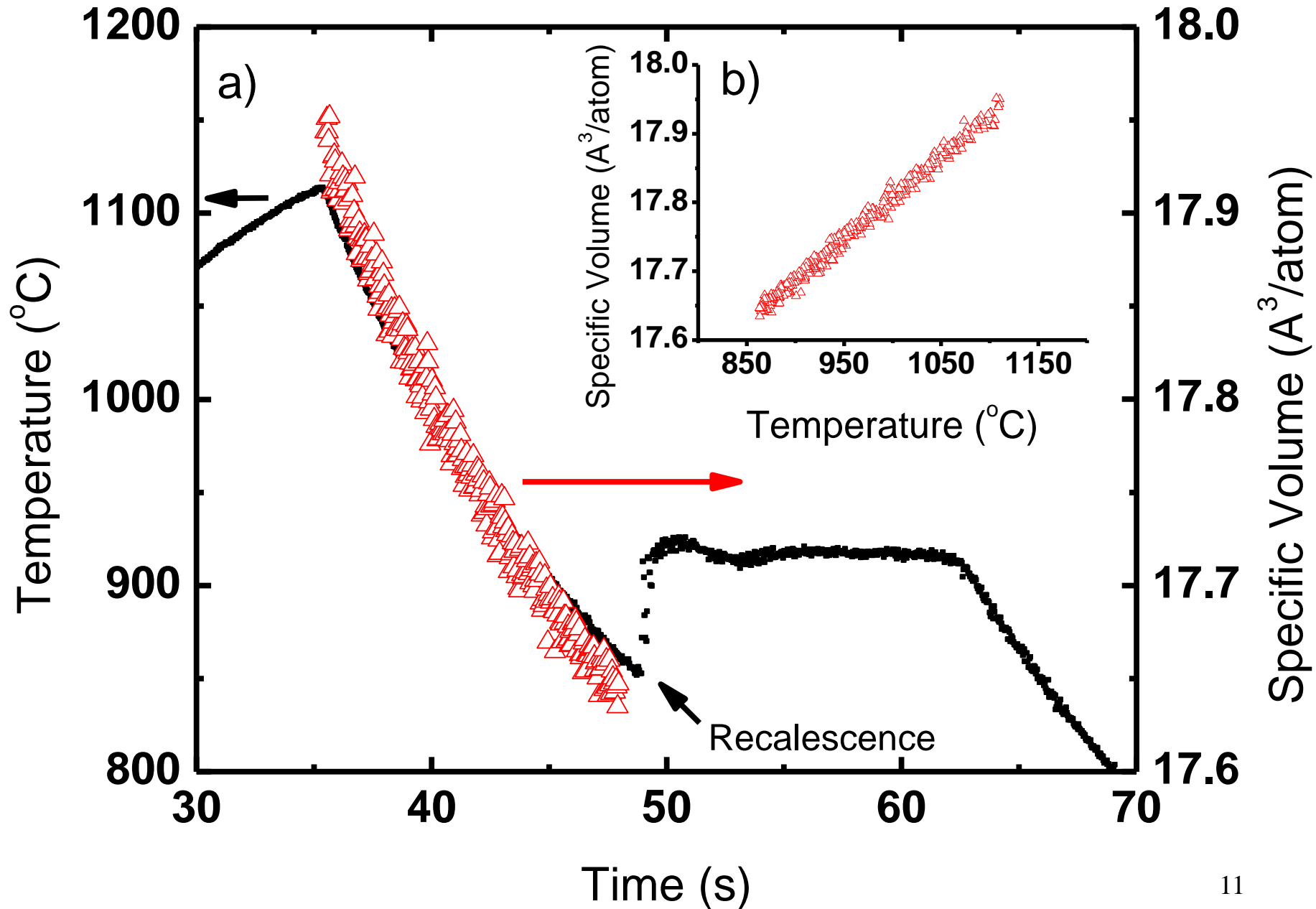


- Density measured from video of sample shadow
- Data processed to identify edges of 2D silhouette (sub-pixel resolution)
- Assume symmetry around vertical axis - calculate volume

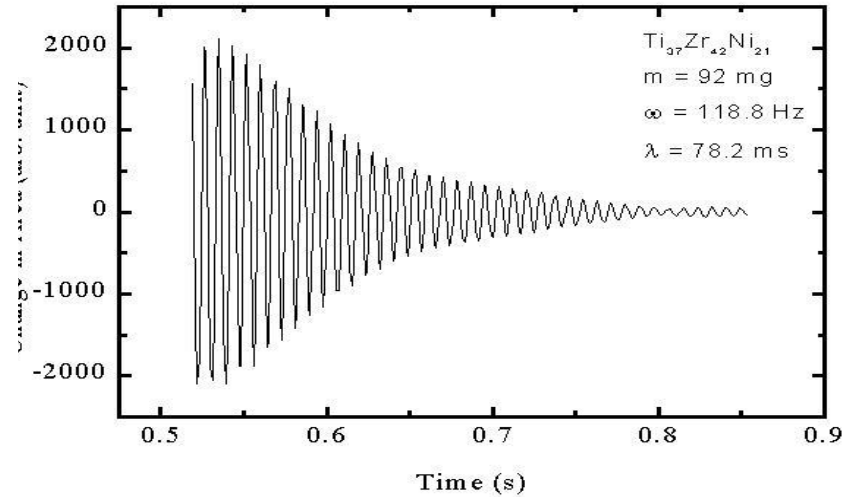
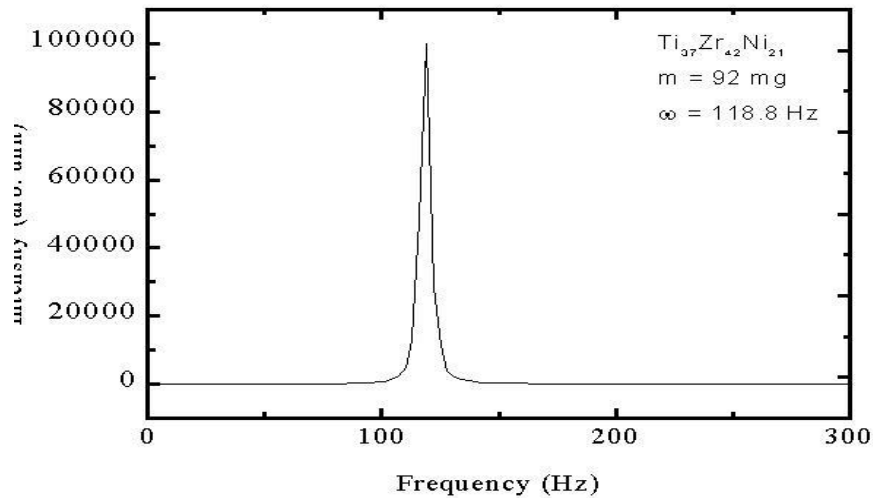
Developed from R. C. Bradshaw, D. P. Schmidt, J. R. Rogers, K. F. Kelton, R. W. Hyers, *Rev. Sci. Instrum.* 76: 12 125108 (2005).



# Characteristic Processing Cycle for $\text{Cu}_{54}\text{Zr}_{46}$



# Viscosity Measurements



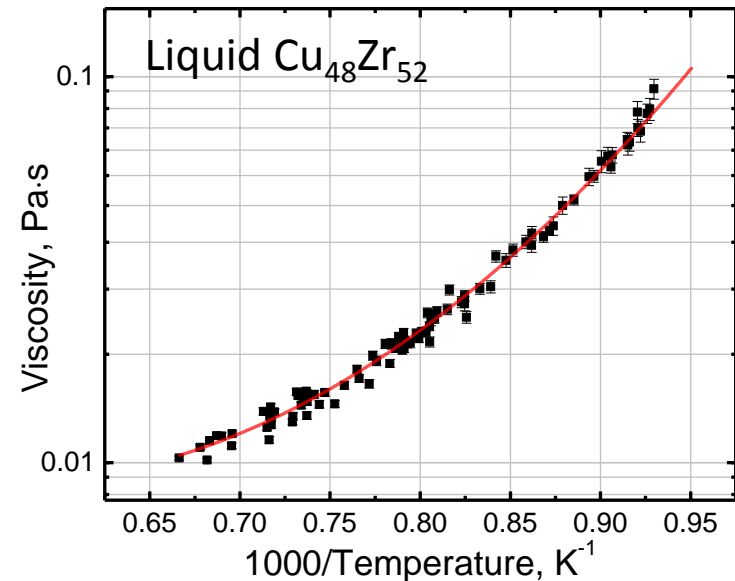
$$R = R_0(1 + \delta \cos(\omega t) e^{-\lambda t})$$

$$\omega_l = \sqrt{\frac{l(l-1)(l+2)\sigma}{\rho R_o^3}}$$

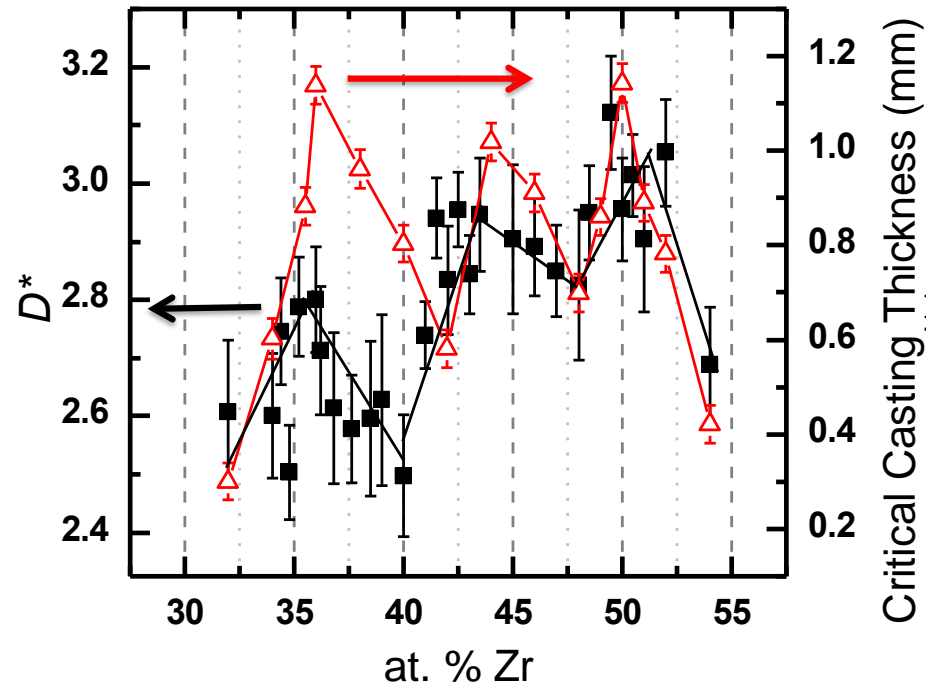
Rayleigh  
(1879)

$$\tau_l = \frac{\rho R_o^2}{(l-1)(2l+1)\eta}$$

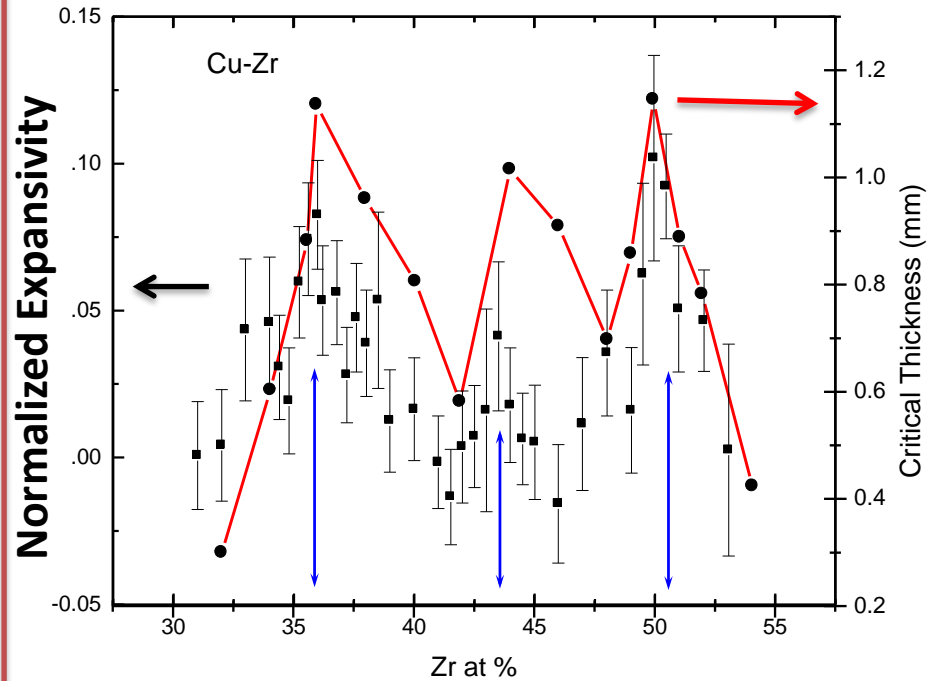
Lamb  
(1881)



# Fragility and Glass Formability



J. C. Bendert and K. F. Kelton,  
*J. Non-Cryst. Solids*, **376**, 205-208, (2013)

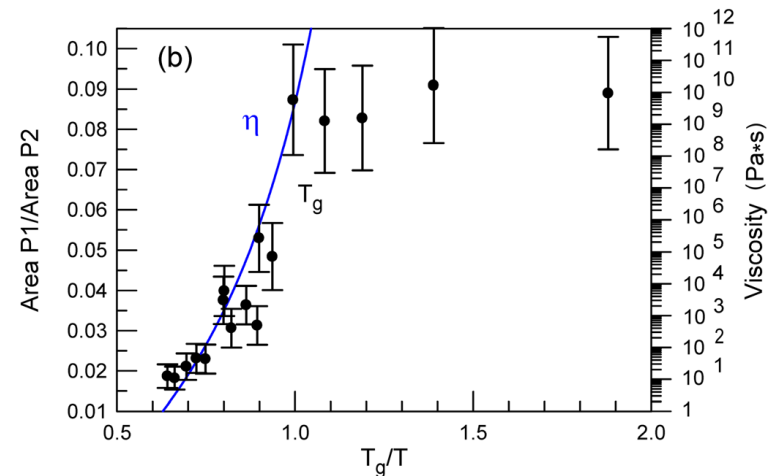
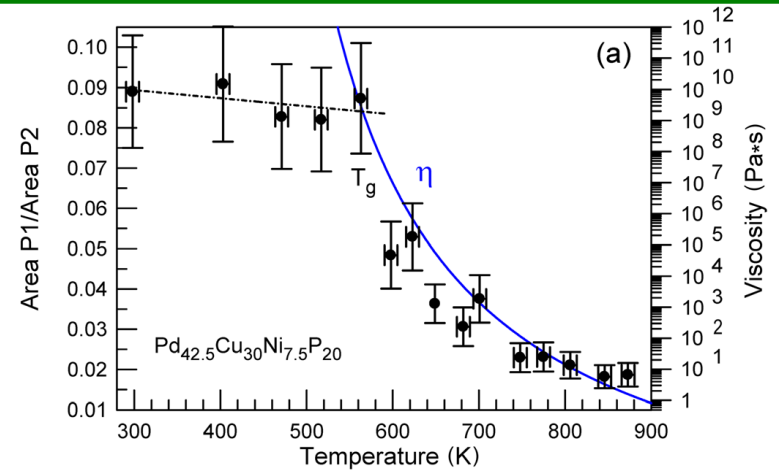
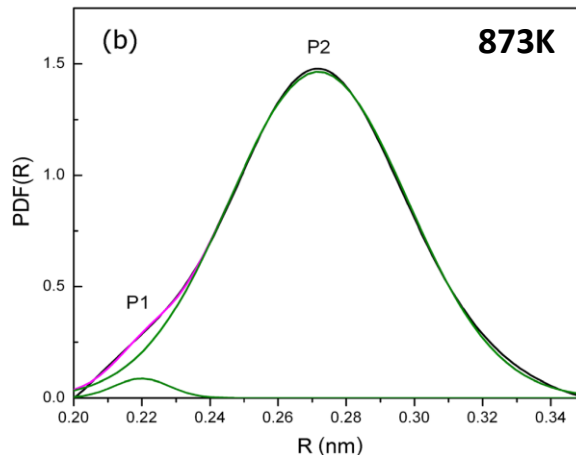
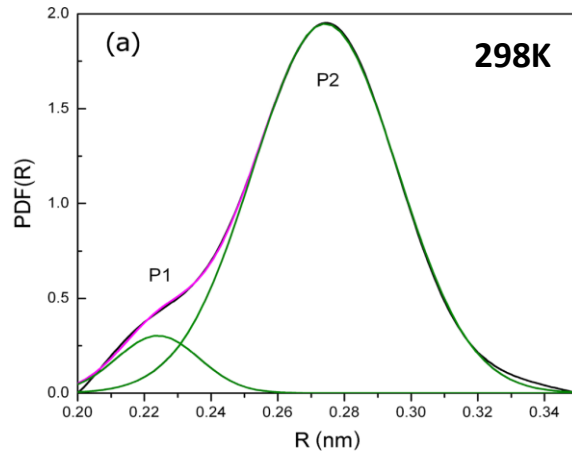


J. C. Bendert, A. K. Gangopadhyay, N. A. Mauro,  
 and K. F. Kelton, *PRL*, **109** (18), 185901 (2012).

$$h = h_0 \exp\left(\frac{D^* T_0}{T - T_0}\right)$$

- Larger expansivity corresponds to larger  $D^*$  .... stronger liquids
- Two components to expansivity in liquids
  - Anharmonic potential & Structural ordering
- Larger expansivity suggests faster ordering rate

# Chemical Ordering in Pd<sub>42.5</sub>Cu<sub>30</sub>Ni<sub>7.5</sub>P<sub>20</sub> BMG Liquid



- Chemical ordering around P (Ni-P and Cu-P bonds increase) on cooling to  $T_g$  (increased magnitude of P1 over P2)
- Viscosity follows the P1/P2 area ratio with supercooling. Suggests -
  - Viscosity (and directly related to chemical ordering in the liquid (clusters))
  - Maybe a structural connection with fragility?

(from D. V. Louzguine-Luzgin *et. al.*, J. Appl. Phys., **110**, 043519 (2011))

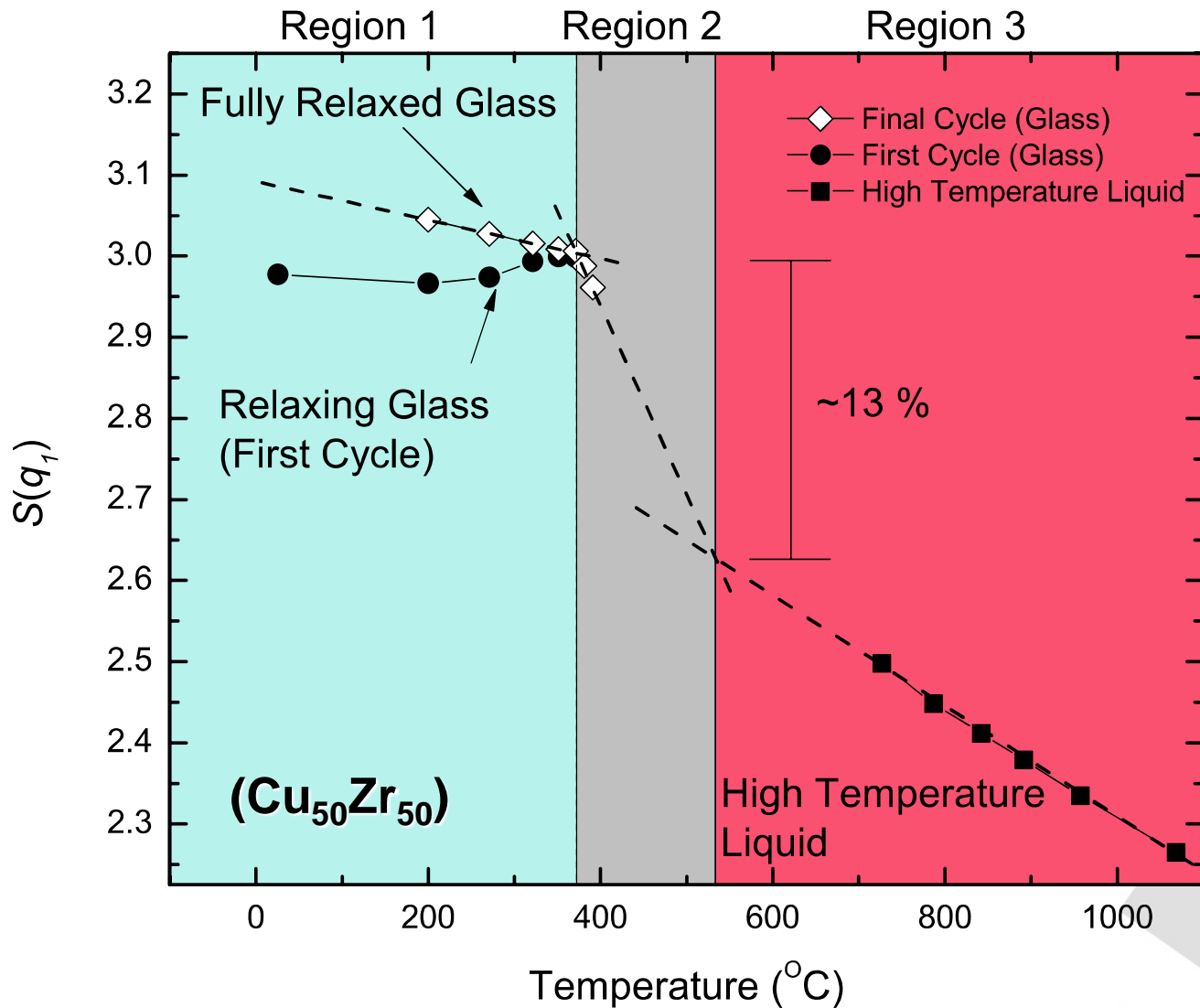
***Thermal Expansivity  
is a Measure of Structural  
Ordering***

**-**

***Should be Observable  
Directly In X-Ray Scattering  
Studies -  $S(q)$***

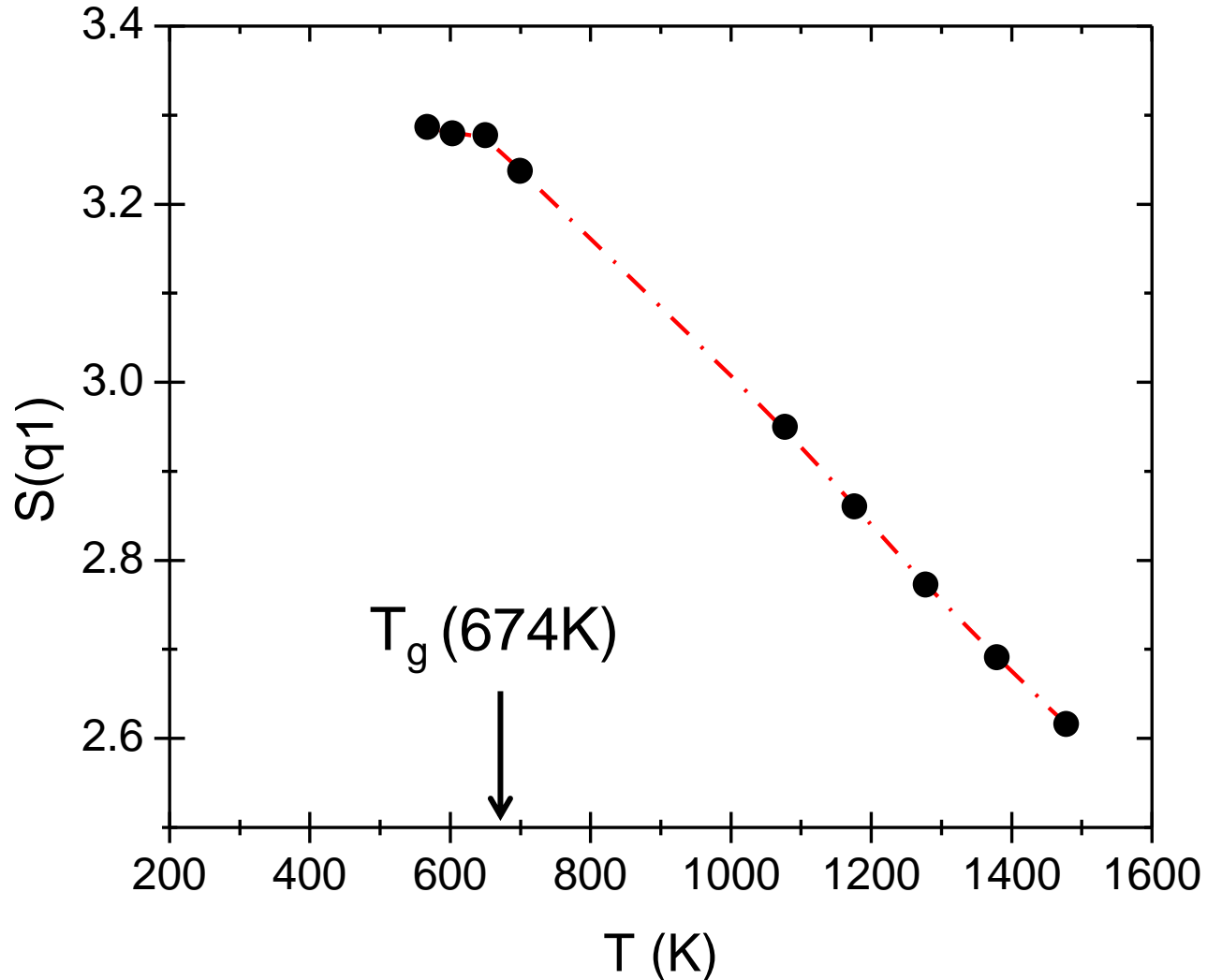
# Structural Ordering

## - Change in First Peak Height in $S(q)$ -



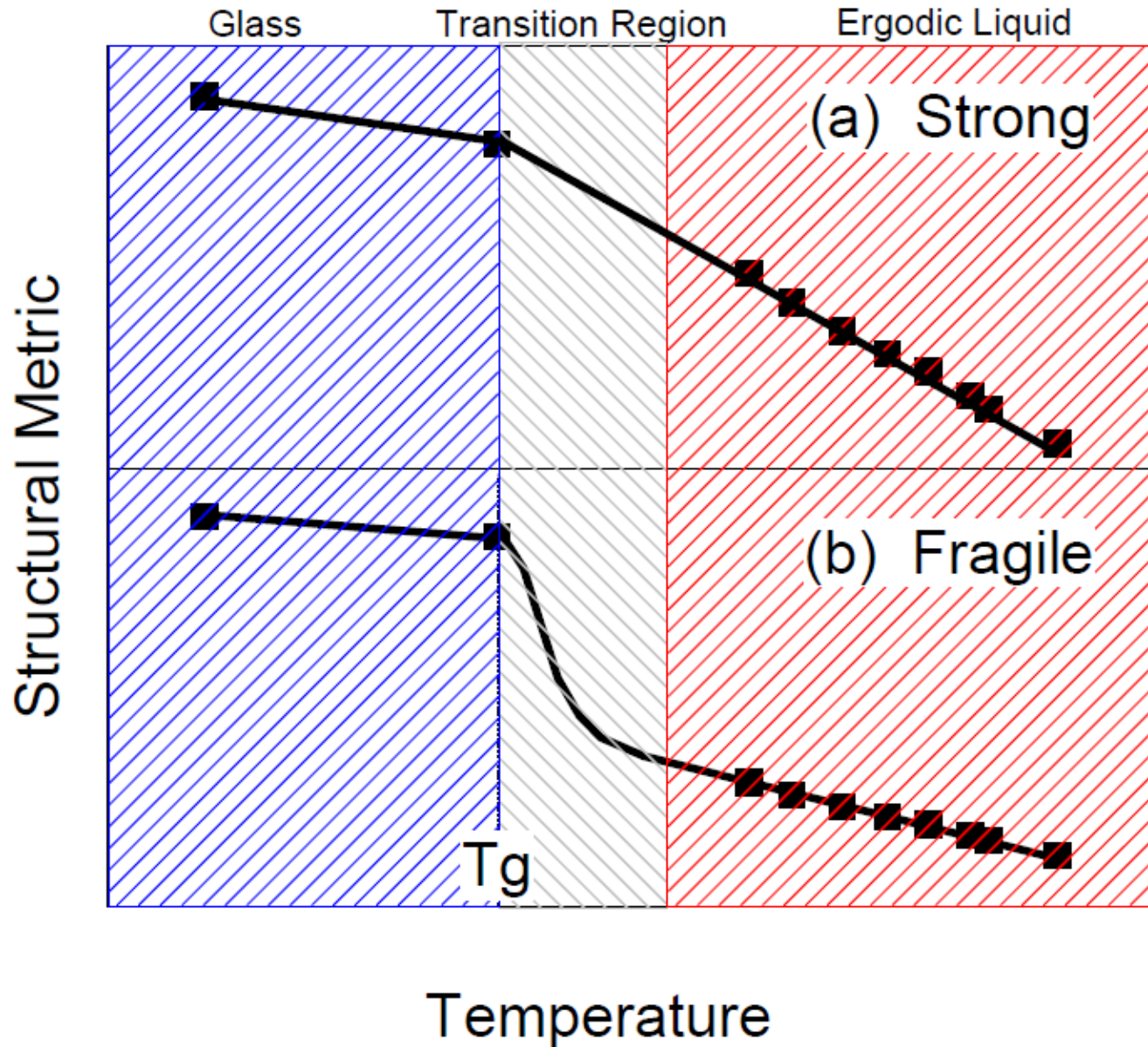


**Vit 106a ( $\text{Zr}_{58.5}\text{Cu}_{15.6}\text{Ni}_{12.8}\text{Al}_{10.3}\text{Nb}_{2.8}$ ) –  
strong liquid and good glass former**



Continuous and gradual change with decreasing temperature

# Fragility Inferred from Structural Data



*Is icosahedral Ordering  
Linked to  
Fragility?*

# Structural Information from S(q) Data

## Reverse Monte Carlo (McGreevy) -- Topological analysis

### Bond Orientational Order Parameter Analysis (Steinhardt *et al.*)

- Calculate bond angles ( $\theta$  and  $\varphi$ ) between atom at center of cluster and vertex atoms
- Express as average order parameter in terms of spherical harmonics

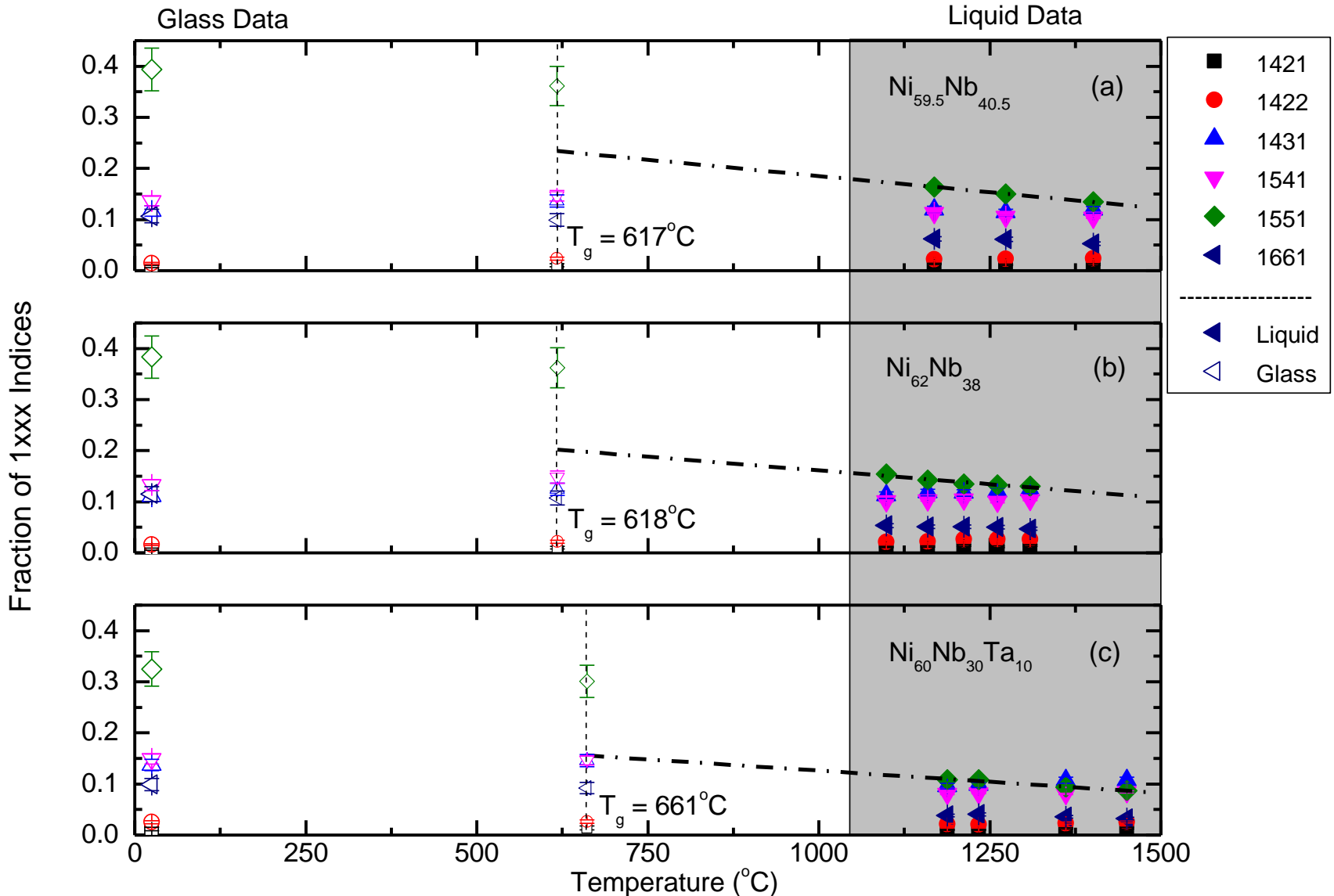
$$Q_l = \frac{4\pi}{2l+1} \frac{1}{N_b} \sum_{m=-l}^l |\overline{Q}_{lm}|^2 \quad \text{where} \quad \overline{Q}_{lm} = \frac{1}{N_b} \sum_{\text{bonds}} Y_{lm}(\theta(\vec{r}), \varphi(\vec{r}))$$

- Icosahedral order –  $Q_6$

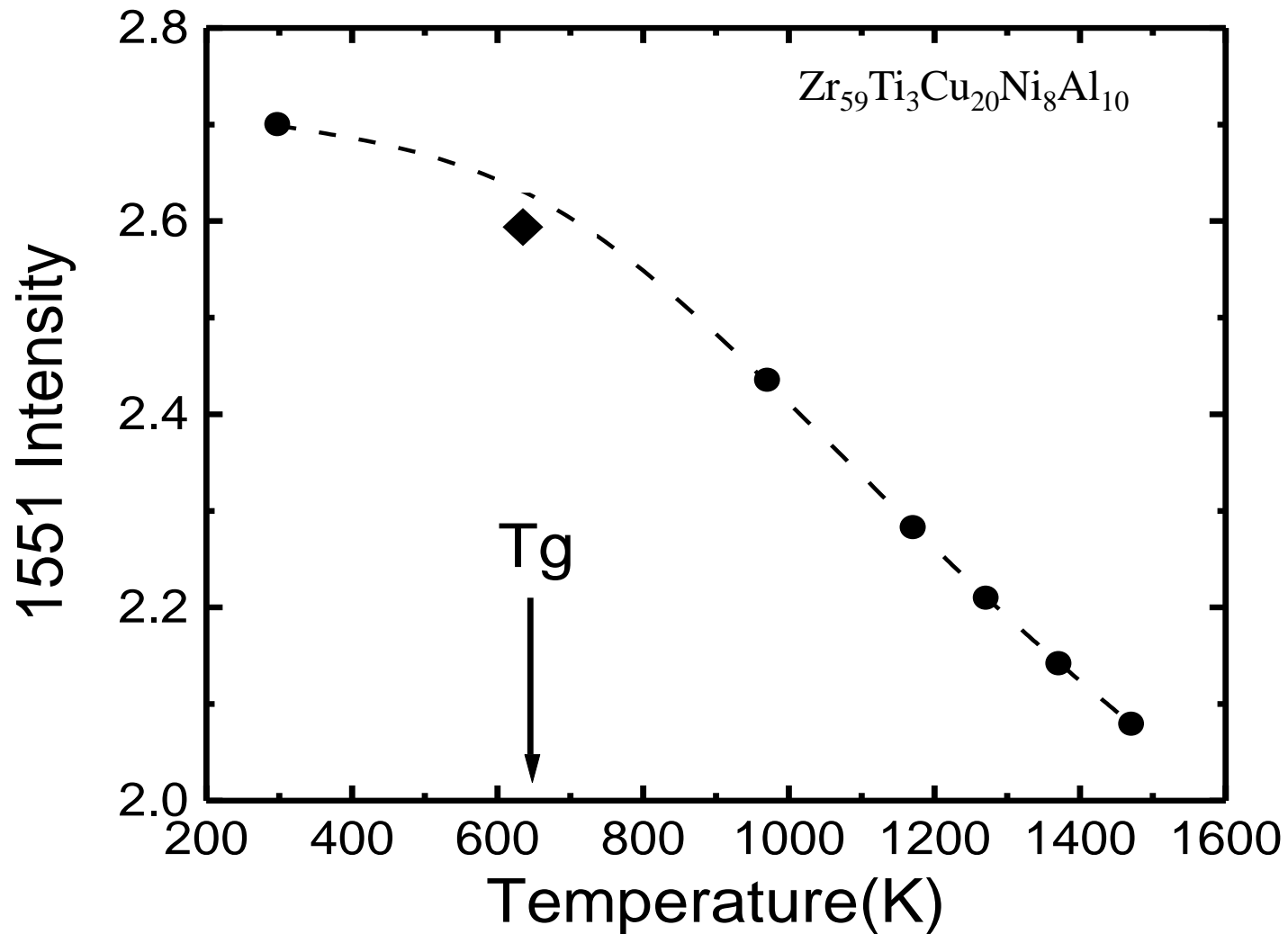
### Honeycutt Andersen Indices (local topology in terms of 4 indices)



# HA 1551 Index for Ni-Nb Glass Forming Liquids



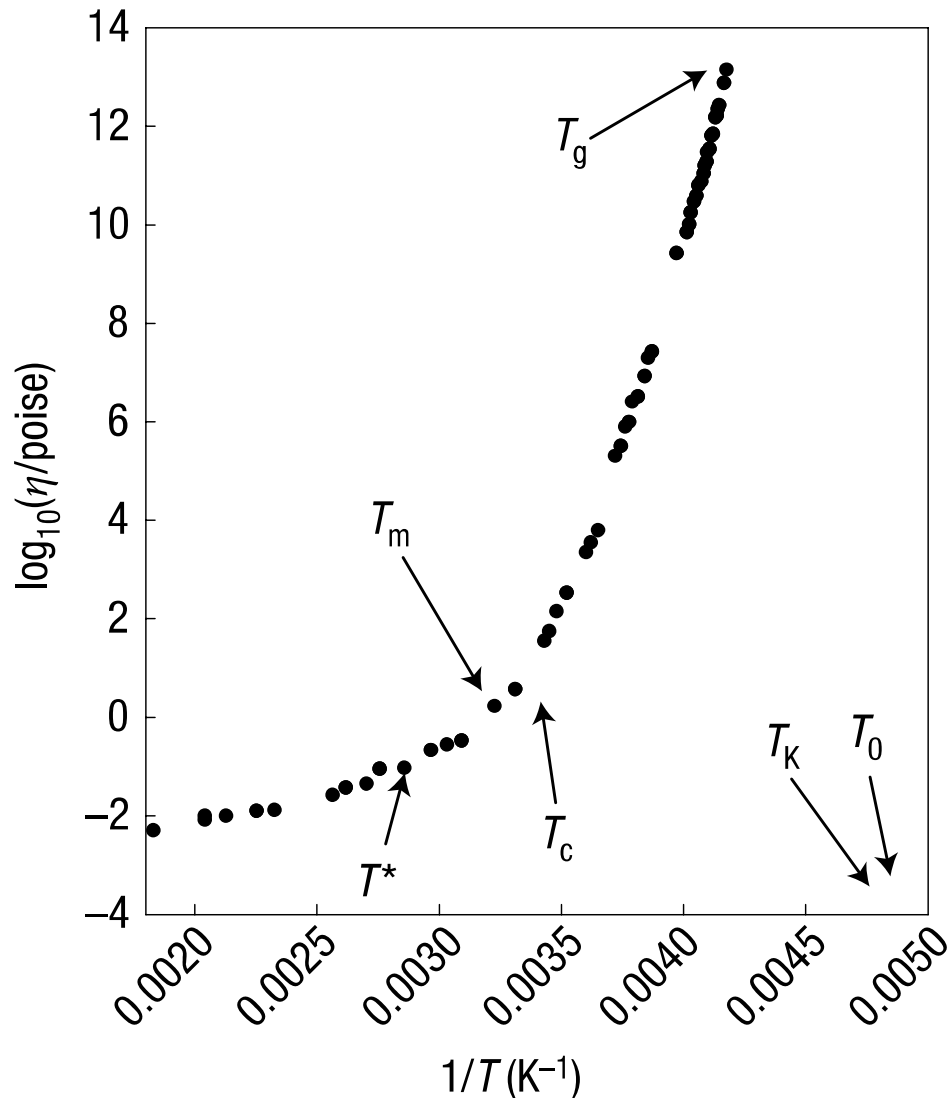
**Strong liquid – good glass former**



# Scaling Temperature

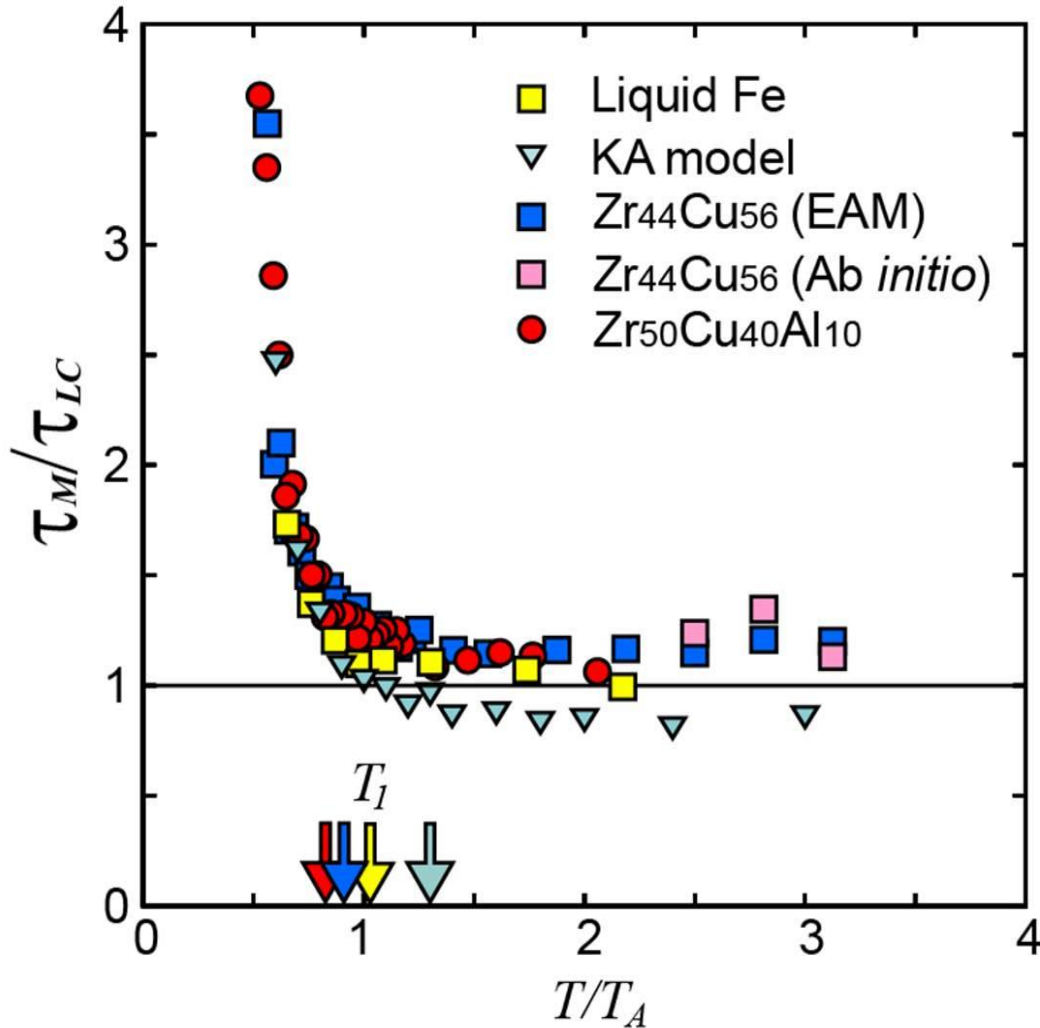
## Possible Fundamental Temperatures

- $T_m$  – Equilibrium melting temperature
- $T^*$  – Crossover temperature for onset of collective behavior in supercooled liquid
- $T_c$  – Mode coupling temperature
- $T_g$  – Glass transition temperature
- $T_K$  – Kauzmann temperature
- $T_0$  – Temperature of unobtainable phase transition



(from S. Kivelson and G. Tarjus, Nature Materials, **7**, 831 (2008))

# $T_A$ – Universal Scaling Temperature?



## High temperature

$$t_M = t_{LC}$$

$t_M$  - Maxwell time ( $h/G_{\neq}$ )

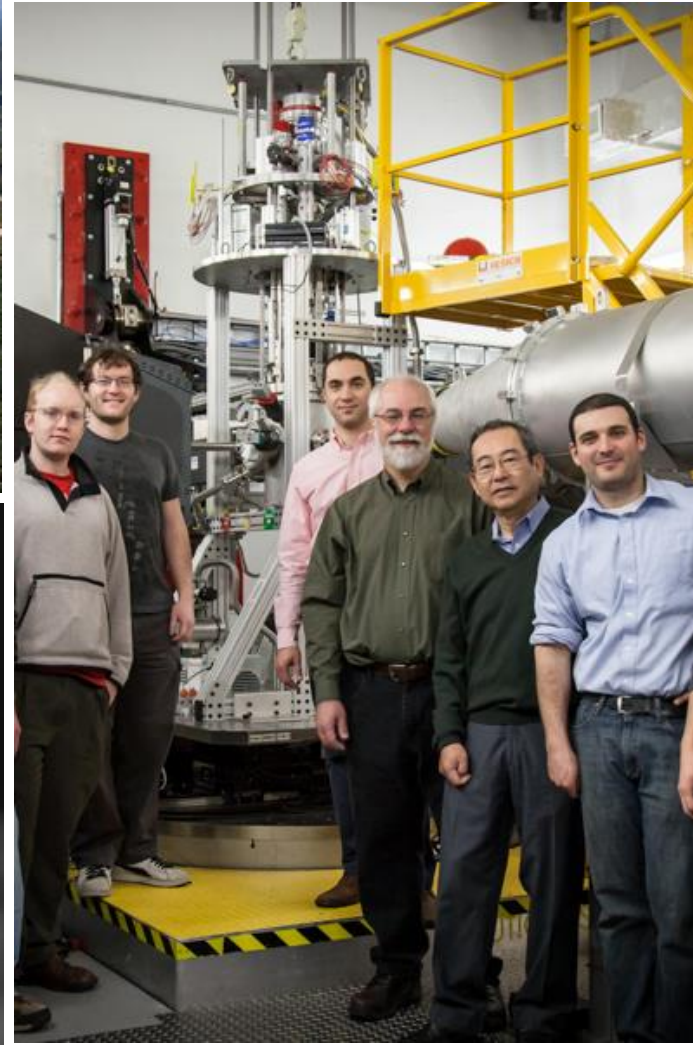
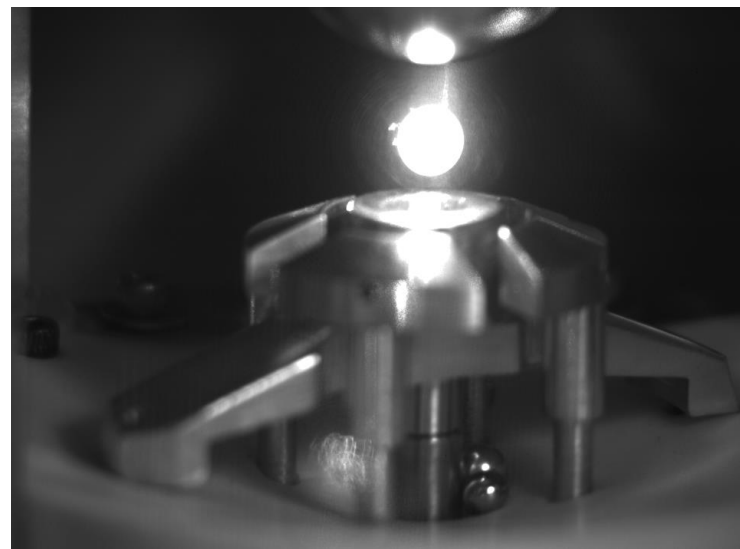
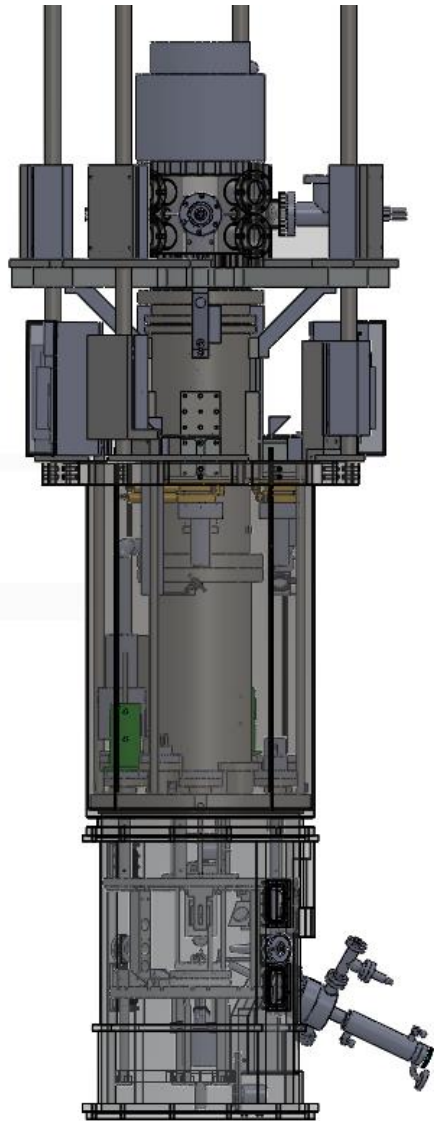
$t_{LC}$  - Time to break bond

- Fe: Johnson potential
- KA: Kob-Andersen potential (Ni<sub>80</sub>P<sub>20</sub>)
- Cu<sub>56</sub>Zr<sub>44</sub>: EAM
- Zr<sub>50</sub>Cu<sub>40</sub>Al<sub>10</sub>: EAM
- Cu<sub>56</sub>Zr<sub>44</sub>: DFT-MD



**We see the same scaling  
in our high  
temperature viscosity  
data**

# Neutron Scattering – NESL (Electrostatic Levitation)



# Conclusions

- Icosahedral short range ordering does not necessarily help glass formability – might enhance nucleation and growth of quasicrystal phases.
- Fragility is correlated with liquid structure
  - Stronger liquids order gradually from high temperatures to the glass transition temperature,  $T_g$ .
  - Fragile liquids order slowly at high temperature, accelerate ordering near  $T_g$ .
- Universal scaling temperature for viscosity
  - Onset of cooperativity
  - Critical slowing down of glass transition anticipated in high temperature liquid

# Acknowledgements

## **Washington University**

- Nicholas Mauro – Structure Factor Measurements
- Matthew Blodgett – Viscosity Measurements
- Zohar Nussinov

## **University of Tennessee**

- Takeshi Egami



*Happy Birthday*



*Austen*